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Variable properties of solid solutions of cerium-samarium tungstates Ce_(2-x)Sm_x(WO₄)₃ (x≤0.3)

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Abstract—A solid solution Ce_(2-x)Sm_x(WO₄)₃ with x≤0.3 was synthesized via a coprecipitation route followed by thermal treatment at 1000 °C. The polycrystalline compounds were characterized by X-ray diffraction, scanning electron microscopy and photoluminescence experiments. Using Rietveld procedure, the decrease of cell parameters showed that a solid solution was obtained. A comparative analysis of diffraction profiles allowed determining structural distortions increasing with composition x. Photoluminescence (PL) experiments were performed under UV and X-ray excitations. The variations with composition x of PL emission intensities, corresponding to complex charge transfers and quenching effects, were interpreted in terms of transitions proper to Ce³⁺ and Sm³⁺ cations, accompanied by additional transitions probably due to oxygen vacancies and other point defects generated by substitution and synthesis conditions. The CIE chromaticity diagrams showed variable colors located in the orange-red range.

Keywords— samarium cerium tungstates, chemical substitution, structural distortions, photoluminescence

I- INTRODUCTION

The tungstates AWO₄ and molybdates AMoO₄ with A = Ca, Sr, Ba, Pb, Cd... were extensively studied for their potential applications: luminescence for detection, low cost lighting, photocatalysis [1]. The synthesis conditions or thermal treatments played a prominent role in the observed final properties. The most common crystalline forms are of scheelite and wolframite types. Recently, various series of solid solutions of photoluminescent materials were synthesized and studied in our laboratories: Ca_(1-x)Cd_xWO₄ [2], Sr_(1-x)Pb_xWO₄ [3], Sr_(1-x)Pb_xMoO₄ [4], Ba_(1-x)Pb_xWO₄ [5]. In each of these series of scheelite solid solutions, the structural, electrical, vibrational and photoluminescence (PL) properties were studied as a function of composition x. The effect of disorder was observed through perturbations of these properties: an increase in photoluminescent emissions was observed for certain intermediate compositions x. Tungstates based compounds are also considered as potential excellent matrix for rare earth (RE) doping to develop luminescent materials [6]. Rare earth tungstates have attracted much attention from researchers because of their well-known photoluminescence emissions [7,8] applicable to scintillators, electro-optical devices, thermoluminescence, stimulated Raman scattering,

catalysis [9]. In the field of luminescence, scheelite tungstates are much interesting because of their self-luminescence character [10]. This family has a self-luminescence in a visible spectrum, it could be divided into three regions: blue, green and red. In the literature, the researchers attributed the emission in the blue region to the charge transfer within the group of transition $(\text{WO}_4)^{2-}$ [11,12]; however, to date, the origins of the green and red luminescence emissions are not definitely interpreted.

Presently, we have focused our attention on new solid solutions based on a matrix $\text{Ce}_2(\text{WO}_4)_3$ in which Sm^{3+} ions substitute for Ce^{3+} ions. The $\text{Ce}_2(\text{WO}_4)_3$ phase was synthesized first by T. Greesling et al. [13]. The authors prepared single crystals of $\text{Ce}_2(\text{WO}_4)_3$ by crystallization from molten mixtures of WO_3 and CeO_2 . The crystal structure was determined by X-ray diffraction on single crystal: the symmetry was monoclinic with space group C2/c and cell parameters a (\AA) = 7.813(4), b (\AA) = 11.720(2), c (\AA) = 11.586(3), β = 109.36(3) ° and with 4 chemical units ($Z = 4$). The structure of $\text{Ce}_2(\text{WO}_4)_3$ was characterized by WO_4 tetrahedra, WO_5 square pyramids and distorted CeO_8 polyhedra. Recently, Anna M. Kaczmarek et al. [14] synthesized the $\text{Ce}_2(\text{WO}_4)_3$ phase by using a hydrothermal method. The authors demonstrated the influence of the presence/absence of the surfactant on the morphology and the photoluminescence properties of Tb^{3+} in the $\text{Ce}_2(\text{WO}_4)_3$ matrix. Other authors, Bai Sun et al. [15] reported a simple hydrothermal way to elaborate the $\text{Ce}_2(\text{WO}_4)_3$ phase: the goal was to fabricate a resistive switching memory device with $\text{Ag}/\text{Ce}_2\text{W}_3\text{O}_{12}/\text{fluorine-doped tin oxide (FTO)}$ structure. Moreover, they demonstrated the effect of visible-light illumination on the device.

The interest of doping $\text{Ce}_2(\text{WO}_4)_3$ by Sm^{3+} cations is multiple: magnetic properties, luminescence of both Ce^{3+} and Sm^{3+} cations combined with charge transfers in $\text{WO}_4 / \text{WO}_5$ groups. This system was never investigated with the objective of determining the interactions between diversified emission centers, which is essential to develop tunable systems of detection or low-cost lighting (Light Emitting Diode, LED).

Wenyan Zhang et al. recently studied the $\text{Sm}_2(\text{WO}_4)_3$ compound [16] presenting the same monoclinic structure and space group as cerium tungstate. These authors demonstrated that, in the $\text{Sm}_2(\text{WO}_4)_3$ matrix, the WO_4^{2-} group could transfer the near UV light energy to the Sm^{3+} ions, emitting blue, green and red light. This property would allow these materials to act as UV absorbers for solar cells and can be useful for the development of white-light LEDs. In parallel, researchers are also attracted by the Ce^{3+} doped materials (with low concentration of dopant <1%) because of their low cost, their spin and their parity which allows 4f - 5d optical transitions characterized by fast-radiative lifetime of about 10 to 50 ns, which is very desirable for scintillators, diodes etc. [17].

In this study, the distant objective would be to determine the role of PL emitting centers in the variations of PL intensities, and to determine their potential applications as materials with tunable photoluminescence.

II- EXPERIMENTAL SECTION

Synthesis. The $\text{Ce}_{(2-x)}\text{Sm}_x(\text{WO}_4)_3$ polycrystalline samples were elaborated via a co-precipitation route, using aqueous solutions of different precursors (cerium nitrate, samarium acetate and sodium tungstate). Stoichiometric amounts of $\text{Ce}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ [Sigma-Aldrich No. 238538, > 99.0%], $\text{C}_6\text{H}_9\text{O}_6\text{Sm} \cdot x\text{H}_2\text{O}$ [Alfa-Aesar No. 14604, > 99.0%] and $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$ [Sigma-Aldrich

No. 72069, > 99.0%] were dissolved in 50 ml of distilled water to individually prepare 2g of $Ce_{(2-x)}Sm_x(WO_4)_3$. Both solutions of $C_6H_9O_6Sm_x \cdot xH_2O$ and $Na_2WO_4 \cdot 2H_2O$ were added dropwise into the cerium nitrate solution maintaining constant stirring. Yellow precipitates formed after pH adjustment at room temperature. These precipitates were washed three times with distilled water and ethanol. A thermal pretreatment at 80 °C for 24 h was carried out. Finally, each precursor sample was compressed and a thermal treatment at 1000 °C for 4 h was applied, giving rise to $Ce_{(2-x)}Sm_x(WO_4)_3$ samarium cerium tungstate compound. **Fig. 1** describes the preparation process of precursor.

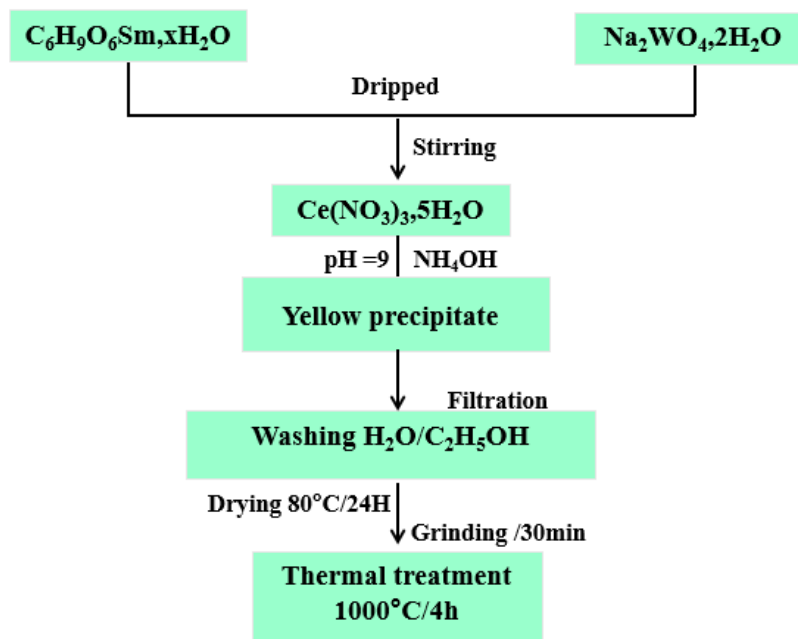


Fig. 1: Synthesis of $Ce_{2-x}Sm_x(WO_4)_3$ polycrystalline samples by coprecipitation method followed by thermal treatment.

Characterizations. X-ray diffraction (XRD). The XRD patterns were acquired with an EMPYREAN Panalytical diffractometer, equipped with a copper X-ray source (Cu $K\alpha_1$ - $K\alpha_2$ doublet: $\lambda = 1.540/1.544 \text{ \AA}$), operating at $V = 45 \text{ kV}$ and $I = 35 \text{ mA}$, with a Ni filter eliminating the $K\beta$ radiation and a rotating sample holder. The diffractometer was equipped with a Pixcel-1D-Detector. The XRD analysis was carried out using the classical θ - 2θ configuration, in continuous mode, with a step size of $0,0032826^\circ$, a scan speed of $0,001394^\circ/s$.

Rietveld analysis using the classical Fullprof software [18] was performed to determine the cell parameters and evaluate the crystallization levels through profile analyses. The initial crystal structure data of Gressling et al. in 1995 [13] were used to refine the as-obtained structures.

Scanning electron microscopy (SEM) analyses. SEM analyses were carried out to characterize the morphology and the local composition of the materials. Images were obtained with a SUPRA 40 VP COLONNE GEMINI ZEISS scanning electron microscope (SEM).

Photoluminescence analyses. The equipment used to perform the measurements of luminescence under UV was a spectrometer Horiba Jobin-Yvon HR800 LabRam. All samples were in form of powders and the wavelength of excitation of an Ar-ion laser was 364.5 nm. The power applied to the samples was fixed to 0.005 mW, with an acquisition time set to 5 s. All spectra were recorded in the range [300 - 1200 nm]. For the luminescence under X-ray excitation, the copper X-ray source of the diffractometer Empyrean-Panalytical was used to irradiate the samples and perform luminescence experiments. The nominal emission conditions were 45 kV /35 mA.

III- RESULTS

Structure and morphology. Using the atom coordinates of authors [13], the crystal structure of $\text{Ce}_2(\text{WO}_4)_3$ is shown in **Fig. 2**. It is constituted of WO_4 and WO_5 groups linked to CeO_8 polyhedra.

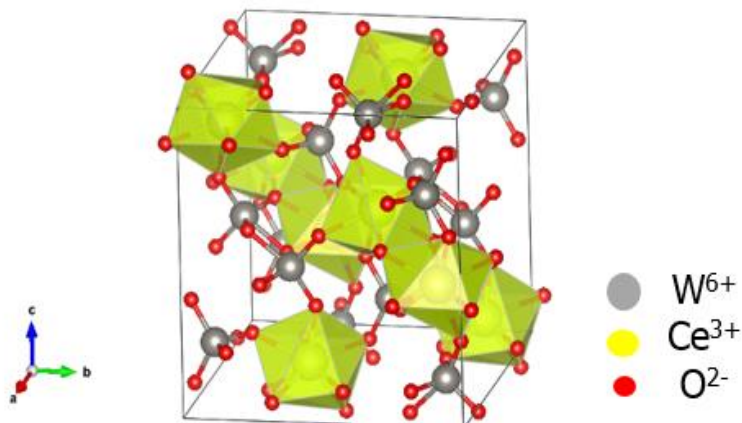


Fig. 2: Crystal structure of $\text{Ce}_2(\text{WO}_4)_3$.

Fig. 3 reports the diffractograms of the $\text{Ce}_{(2-x)}\text{Sm}_x(\text{WO}_4)_3$ system ($0 \leq x \leq 0.3$), obtained after thermal treatment at 1000 °C during 4 h. The analysis of the diffraction profiles shows good crystallization of the synthesized materials with the presence of a minority CeO_2 residual phase. **Fig. 4** shows the enlargement of the angular domain between 24 et 34 ° (2θ).

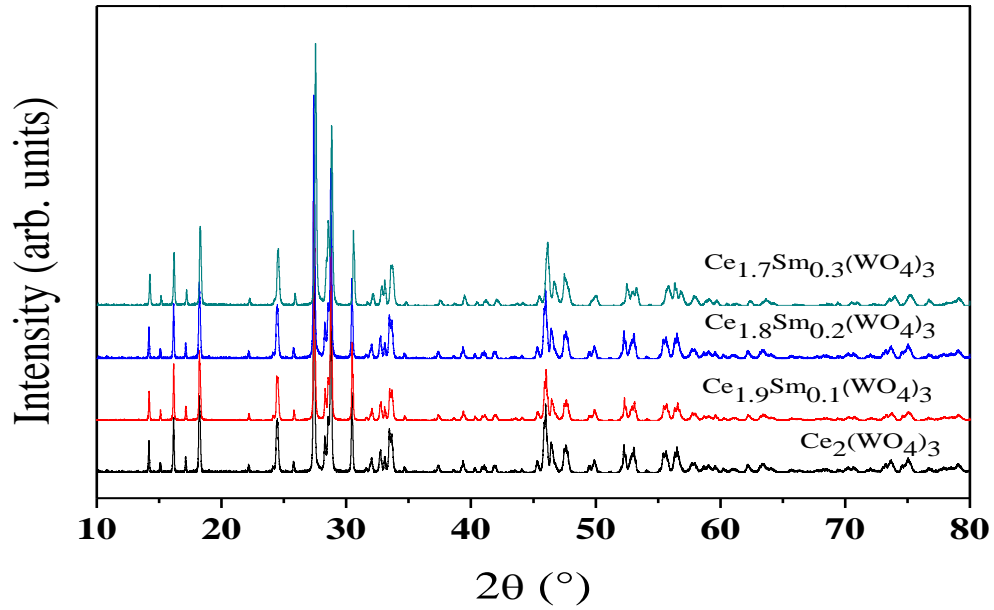


Fig. 3: XRD pattern of $Ce_{2-x}Sm_x(WO_4)_3$ $0 \leq x \leq 0.3$ obtained after thermal treatment at 1000 °C.

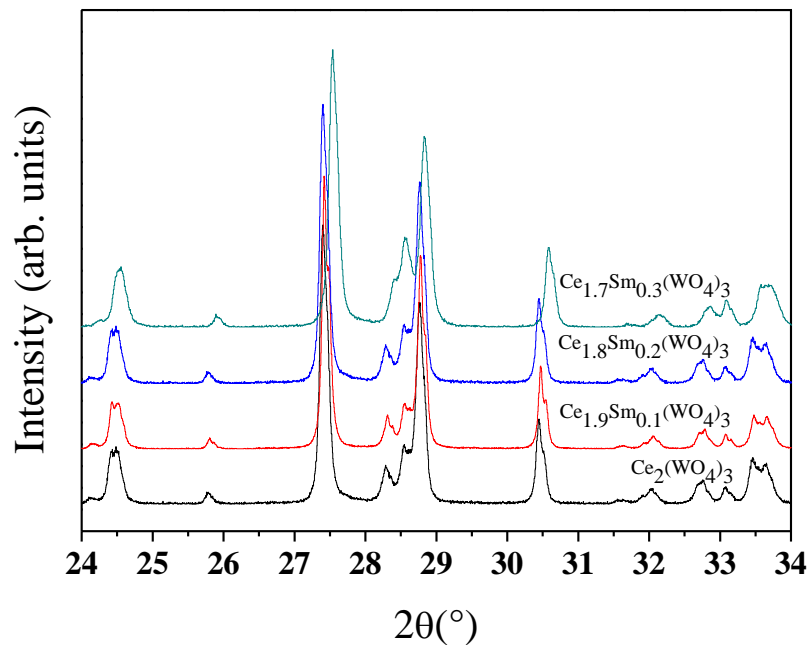


Fig.4: Enlargement of the angular domain between 24 et 34 ° (2θ)

To determine the structural evolution, Rietveld refinement procedure was applied. The atom coordinates given in **annex** were fixed during the refinements: this was necessary because of the limited data delivered by powder diffraction and the high number of variable atom coordinates. Using the XRD analysis of the $Ce_2(WO_4)_3$ ($x=0$) considered as a standard, the refinements consisted in determining the cell parameters and the variations of full width at half maximum (FWHM) as a function of composition x . **Table 1** reports the cell parameters and volumes of the $Ce_{(2-x)}Sm_x(WO_4)_3$ samples thermally treated at 1000°C. **Fig. 5** shows the variations of cell

volumes V as a function of x . A quasi-linear correlation is observed with the volume of the $\text{Sm}_2(\text{WO}_4)_3$ isomorphous [16].

Table 1: Rietveld structure refinement of the $\text{Ce}_{(2-x)}\text{Sm}_x(\text{WO}_4)_3$ samples thermally treated at 1000 °C. Cell parameters, volume, density, profile parameters, reliability factors.

Composition x	0	0.1	0.2	0.3
a (10^{-10} m)	7.81205(4)	7.80763(4)	7.80034(4)	7.78022(4)
b (10^{-10} m)	11.72915(3)	11.72233(4)	11.70735(4)	11.68704(4)
c (10^{-10} m)	11.6144(3)	11.61164(4)	11.60374(4)	11.60053(4)
β (°)	109.337	109.330	109.354	109.308
V : Cell Volume (10^{-30} m ³)	1004.17(3)	1003.23(4)	999.78(4)	995.47(4)
Calculated density (g.cm ⁻³)	6.772	6.788	6.817	6.850
Profile parameters				
U	0.117410	0.141274	0.172059	0.211064
V	-0.019543	-0.021653	-0.024308	-0.028276
W	0.007883	0.004235	0.006833	0.008855
Shape Parameter	0.536650			
Reliability factors				
R_b (%)	4.90	5.01	4.33	5.48
R_f (%)	2.60	2.61	2.17	2.99
R_{exp} (%)	9.97	7.63	8.77	8.49
R_p (%)	13.3	12.4	12.3	12.2
R_{wp} (%)	13.6	12.6	12.1	12.3

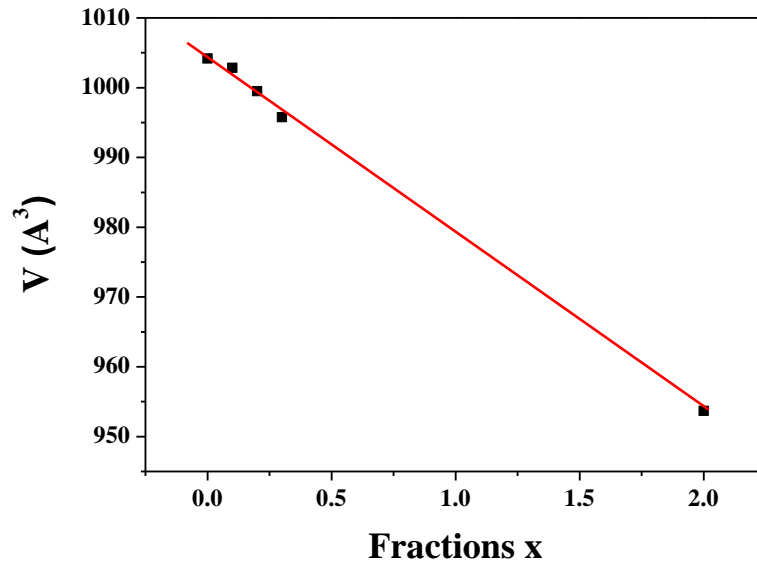


Fig. 5: Variation of volume V (in Å^3) as a function of x of $\text{Ce}_{(2-x)}\text{Sm}_x(\text{WO}_4)_3$. Volume of $\text{Sm}_2(\text{WO}_4)_3$ for $x=2$ from ref. [16].

Fig.6 shows SEM images of the polycrystalline phases constituting complex particle size distributions. A gradual evolution of morphologies and grain sizes is observed when the composition x increases.

- For the compound CeW ($x=0$), we observe a majority of crystallites of micrometric sizes, faceted smooth and regular, having grain boundaries.
- For the composition of $x=0.1$, large crystallite sizes are observed, with facets containing cavities.
- For compositions 0.2 and 0.3, the crystallites are of varying large sizes ($> 5 \mu\text{m}$) and are covered by thin particles with submicronic sizes.

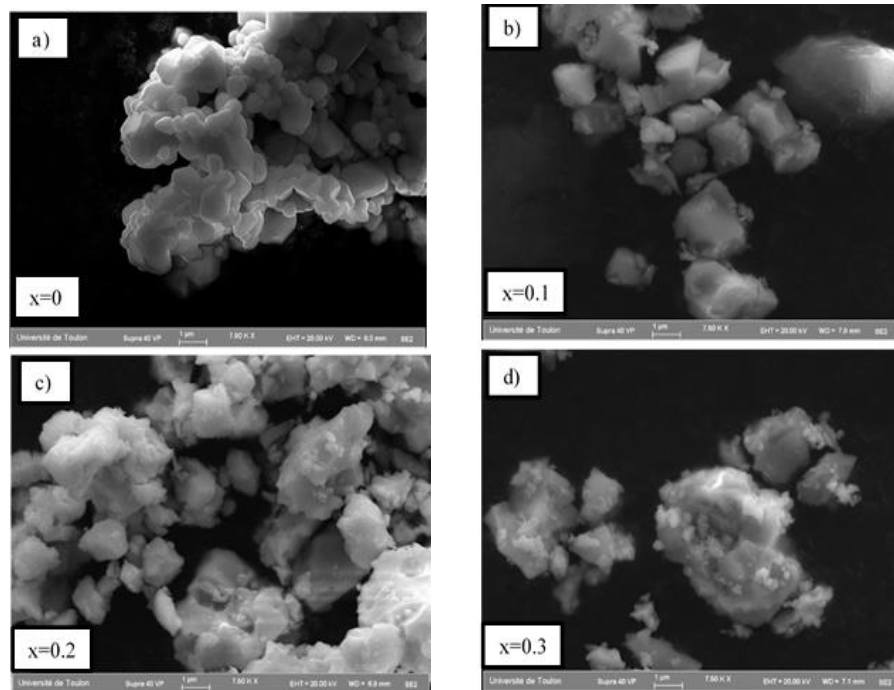


Fig.6. a,b,c,d: SEM images of the $\text{Ce}_{(2-x)}\text{Sm}_x(\text{WO}_4)_3$ samples thermally treated at 1000°C ; (a) $x=0$; (b) $x=0.1$; (c) $x=0.2$; (d) $x=0.3$.

XRD profile analyses. Having regard to the complexity of XRD profiles of these monoclinic structures, where many observed peaks are the superposition of different (h,k,l) Bragg peaks, it was not possible to determine the individual broadening of Bragg peaks in the full diffraction angle range. So, we have used the results of Rietveld refinements to determine individual calculated full width at half maximum (FWHM) for each (h,k,l) Bragg peak.

Considering the sample $x=0$ as a standard, we have determined the modifications of each observed FWHM for $x>0$ and ascribed the additional FWHM increase to substitution effect. The resulting size and/or distortion effects were determined by applying a Williamson Hall test. As a first step, it was necessary to carry out corrections using Lorentzian model as a simplified approach. The corrected FWHMs, noted as $\Delta 2\theta$, were calculated from the difference between the observed FWHM_{obs} for $x>0$ and the value FWHM_{obs} for $x = 0$ for each (h,k,l) reflection:

$$\Delta 2\theta = \text{FWHM}_{\text{obs}}(x) - \text{FWHM}_{\text{obs}}(x=0).$$

Then, the Williamson-Hall test was performed by reporting $\Delta 2\theta$ vs. $\sin\theta$:

$$\Delta 2\theta \cdot \cos\theta = 0.9 \lambda / L + 4\varepsilon \cdot \sin\theta \quad (1)$$

In this expression (1), θ is the Bragg angle, λ is the wavelength (1.54 Å), L is an hypothetical average grain size or coherence length, and ε is the lattice distortion defined as the averaged ratios $\Delta a/a$, $\Delta b/b$ and $\Delta c/c$, or $1/3 \Delta V/V$. The calculations gave insignificant L values (close to 0 with large uncertainties) and only distortion ε values were obtained. This is congruent with our SEM analyses showing that the major population of crystallites had sizes greater than 1 μm . So, we have assumed that only distortion effects must be considered in the X-ray diffraction analyses.

Fig.7 shows the increasing values of distortions $\varepsilon(x)$ which are due to combined effects of different cation sizes of Ce^{3+} and Sm^{3+} , and of defects generated during crystal growth. It should be noted that these values were calculated considering the $x=0$ sample as a standard without any distortion due to substitution. However, it is necessary to note that initial distortions due to crystal growth could not be considered in this approach.

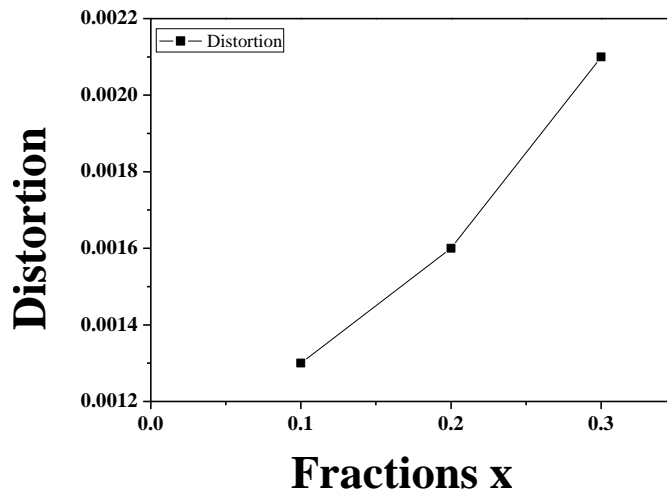


Fig. 7: Variations of lattice distortion ε as a function of composition x .

Luminescence properties. The photoluminescence spectra obtained under X-Ray (45 kV-35 mA) and UV (364.5 nm) excitations are shown in **Figs. 8 a,b,c,d** and **Figs. 9 a,b,c,d** respectively. Emissions of Ce^{3+} and Sm^{3+} observed under UV excitation correspond respectively to the two $[^2\text{Eg}, ^2\text{T}2\text{g}] \rightarrow 2\text{F}_{7/2,5/2}$ ($j = 7/2, 5/2$) transitions and to the four $^4\text{G}_{5/2} \rightarrow ^6\text{H}_j$ (with $j = 5/2, 7/2, 9/2, 11/2$) transitions respectively [19]. The Sm^{3+} emissions are observed in PL experiments under X-ray excitation. We note first that the charge transfer (CT) in WO_4 groups (blue emission) is not observed under X-ray excitation and its presence is questionable in the case of UV excitation. We note also that the luminescence analysis under X-Ray excitation showed the absence of two Ce^{3+} components (497, 524 nm). The two additional emissions at 619 nm and 664 nm cannot be attributed directly to the Ce^{3+} emission center. The luminescence in the 400-1200 nm spectral range is characterized by a broad spectral band [850-1000 nm].

A detailed analysis of the variations of intensities shows that, as x increases:

- The weak Ce^{3+} emission intensities close to 500 nm (2.5 eV), only observed under UV excitation, first increase then decrease;
- The emission intensities at 617-619 and 644-645 nm observed under UV and X-ray excitations are strong, first increase then decrease;
- The emission intensities associated with Sm^{3+} transitions are relatively strong and continuously decrease;

The broad emission spectrum in the 800 to 1000 nm IR range, with a maximum at 900 nm, is observed under both UV and X-ray excitations, however it is strongly increasing in the case of UV excitation.

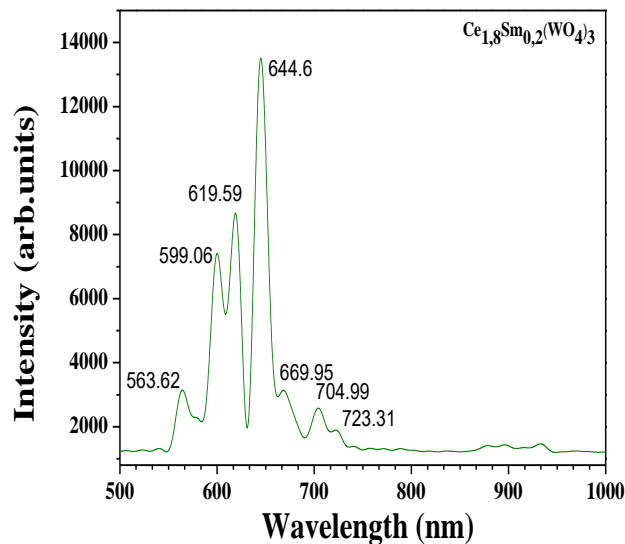
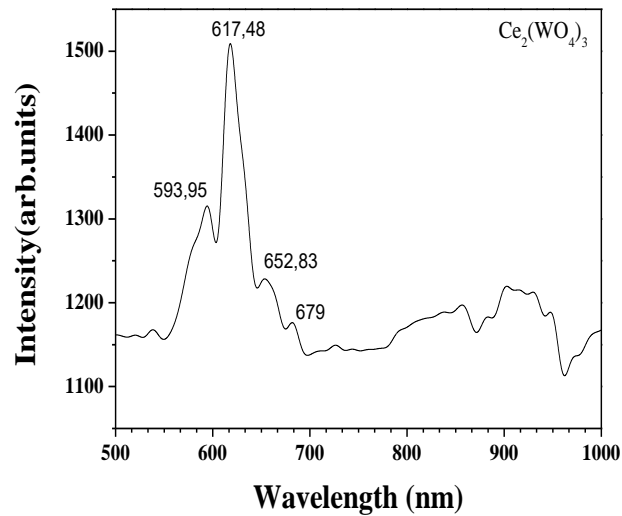


Fig. 8 a,b: Photoluminescence spectra under X-Ray excitation (45 kV-35mA) of $Ce_{2-x}Sm_x(WO_4)_3$ system with x=0, 0.1.

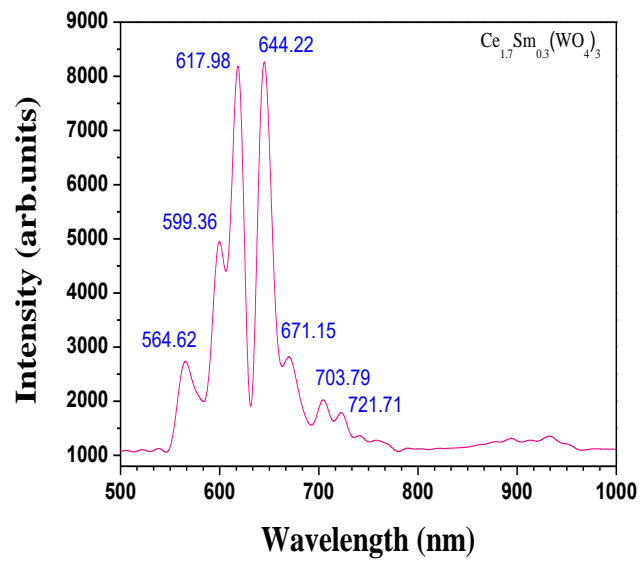
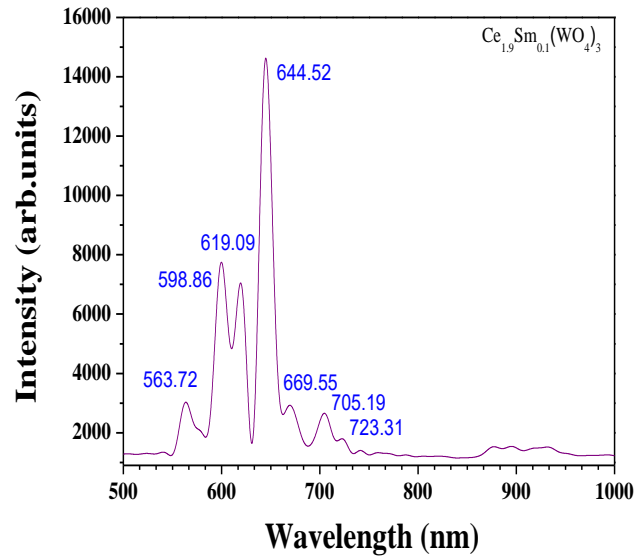


Fig. 8 c,d: Photoluminescence spectra under X-Ray excitation (45 kV-35mA) of $\text{Ce}_{2-x}\text{Sm}_x(\text{WO}_4)_3$ system with $x=0.2, 0.3$.

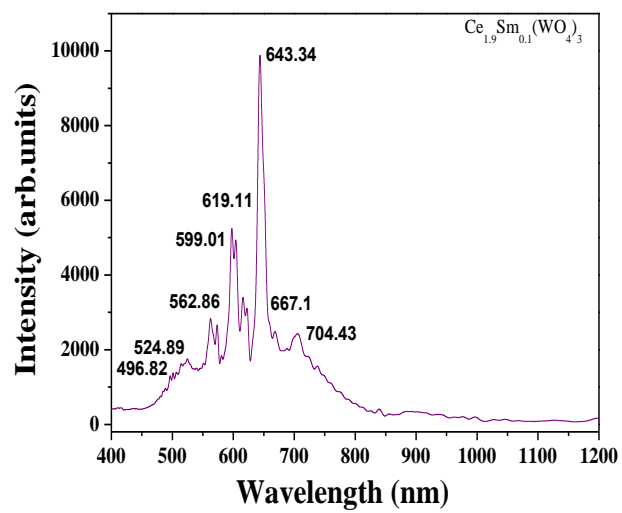
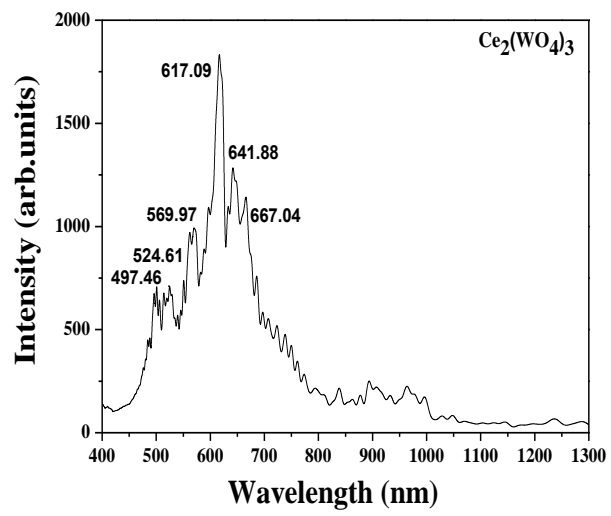


Fig. 9 a,b: Photoluminescence spectra under UV excitation (λ excitation = 364.5 nm) of $\text{Ce}_{2-x}\text{Sm}_x(\text{WO}_4)_3$ system with $x=0, 0.1$.

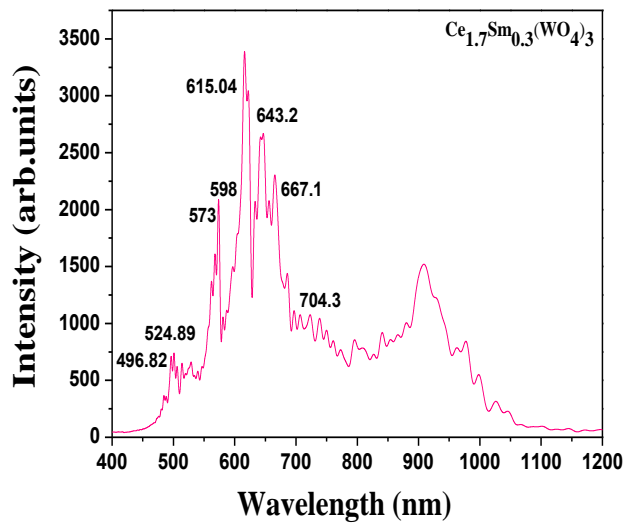
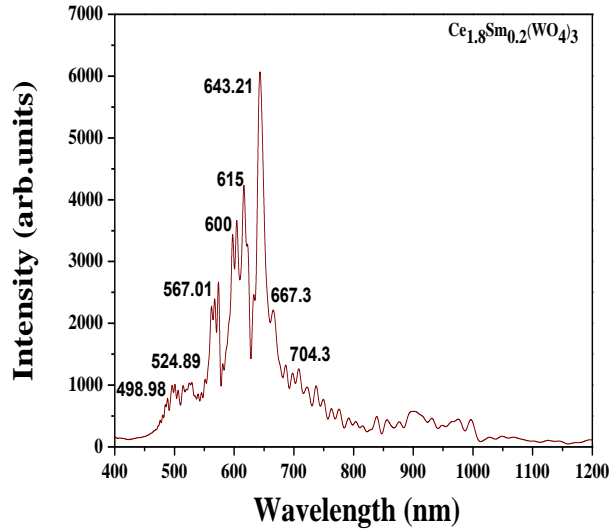


Fig. 9 c,d: Photoluminescence spectra under UV excitation (λ excitation = 364.5 nm) of Ce_{2-x}Sm_x(WO₄)₃ system with x=0.2, 0.3.

Chromaticity diagrams. The colors associated with the various PL spectra are represented in the classical CIE chromaticity diagrams. The CIE coordinates associated with PL emissions under X-ray and UV excitations are reported in **Table 2**.

Table 2: CIE coordinates (x_{CIE} , y_{CIE}) for PL emissions of the $Ce_{2-x}Sm_x(WO_4)_3$ compounds.

X-ray excitation				
Composition	0	0.1	0.2	0.3
x_{CIE}	0.39	0.50	0.50	0.50
y_{CIE}	0.38	0.36	0.35	0.35
UV excitation				
Composition	0	0.1	0.2	0.3
x_{CIE}	0.47	0.49	0.52	0.53
y_{CIE}	0.43	0.41	0.42	0.42

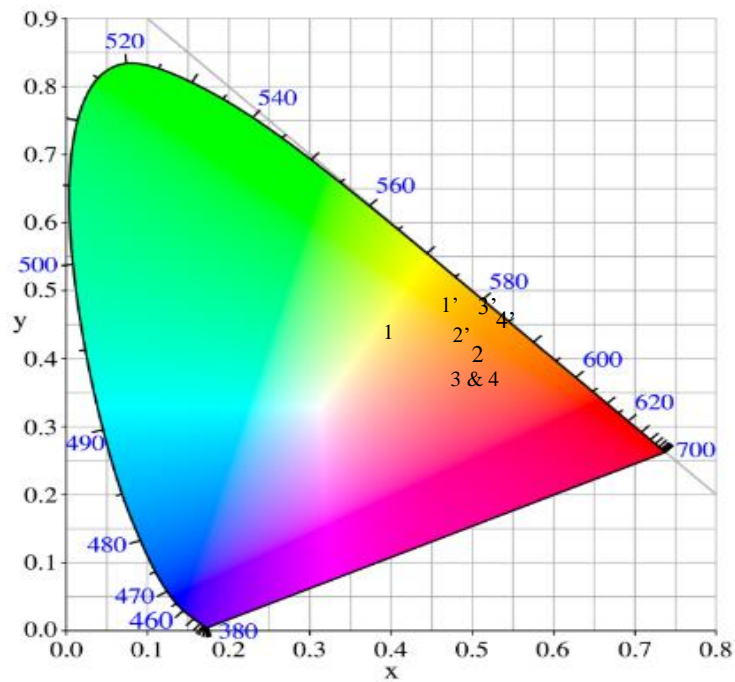


Fig. 10: Chromaticity diagram for $Ce_{2-x}Sm_x(WO_4)_3$ system under both excitations. Under X-ray excitation, points are noted as 1 for $x=0$, 2 for $x=0.1$, 3 for $x=0.2$ and 4 for $x=0.3$. Under UV excitation points are noted as 1' for $x=0$, 2' for $x=0.1$, 3' for $x=0.2$, 4' for $x=0.3$.

IV- DISCUSSION.

In the present work, we managed to form a solid solution of cerium samarium tungstate, in the Sm composition range 0 to 0.3. The quasi-linear decrease of cell parameters shows that

substitution of Ce^{3+} by Sm^{3+} has been successfully completed (Table 1). Using an adapted profile diffraction analysis extracted from Rietveld method, increasing lattice distortions due to increasing composition x were clearly evidenced. This clearly shows that this disordered solid solution is constituted of a disordered distribution of Sm in the lattice, with local bond length variations and with local defects (vacancies and local distortions) likely to play the role of emission centers. The photoluminescence was observed under polychromatic X-ray and monochromatic UV excitations.

It should be recalled that Anna M. Kaczmarek et al [14] in their study of the two $\text{Ce}_2(\text{WO}_4)_3$ and $\text{Ce}_{10}\text{W}_{22}\text{O}_{81}$ phases doped with Tb^{3+} observed some changes in the luminescence spectra depending on morphologies. The authors proposed different energy transfer (ET) mechanisms for the two cerium tungstate materials. In a recent work [20], other authors proposed a complex system of energy levels due to oxygen vacancies. In agreement with their approach, using the classical notation of Kröger and Vink for point defects, we assume that, in the present solid solution, three types of oxygen vacancies neutral or positively charged (noted V^q with $q = 0, 1, 2$) can be formed:

- Neutral V^x oxygen vacancies with 2 trapped electrons occupying an orbital (also noted as F^0 by authors [20])
- Singly charged V° oxygen vacancies with 1 trapped electron (noted as F^+ [20])
- Doubly charged $\text{V}^{\circ\circ}$ oxygen vacancies with 0 electron (noted as F^{++} [20]): this defect can trap one or two electrons, however, it cannot have any excited state.

Under photonic excitation, the V^x and V° defects can reach excited states noted V^{x*} (close to V° or $\text{V}^{\circ\circ}$) and $\text{V}^{\circ*}$ (close to $\text{V}^{\circ\circ}$). The $\text{V}^{\circ\circ}$ cannot be excited (no electron present).

According to these hypotheses, at least four PL transitions could be expected:

- 1 electron in level V^{x*} transiting to level V^x to form V^x ($2e^-$);
- then: V^x ($2e^-$) transiting to holes in O2p valence band;
- 1 electron in level $\text{V}^{\circ*}$ transiting to level V° to form V° ($1e^-$);
- then: V° ($1e^-$) \rightarrow holes in O2p valence band.

Considering the energy gap E_g of the extreme compounds $\text{Ce}_2(\text{WO}_4)_3$ and $\text{Sm}_2(\text{WO}_4)_3$ of about 3.5 and 3.8 eV respectively, all intermediate levels associated with vacancies have energies less than 3 eV.

The two PL emissions at 495 and 525 nm (blue -green emissions) can be attributed to the Ce^{3+} transitions involving the two $^2\text{F}_{5,2,7/2}$ states. The two PL emissions at 619 and 664 nm (orange-red emissions) can be attributed to the two transitions $\text{V}^{x*} \rightarrow \text{V}^x$ and $\text{V}^{\circ*} \rightarrow \text{V}^\circ$. The PL emissions in the IR range (large spectrum between 800 and 1000 nm with maximum at 900 nm) can be attributed to transitions ($\text{V}^x, \text{V}^\circ$) \rightarrow holes in O2p orbitals.

It should be noted that the 2 emissions associated with Ce^{3+} and the 2 additional emissions are constituted of thin bands and could be also associated with typical defect clusters: $\text{W}^{6+} - \text{V}_O^{(q^\circ)} - \text{Ce}^{3+}$ and $\text{Ce}^{3+} - \text{V}_O^{(q^\circ)} - \text{Ce}^{3+}$. Each complex defect can act as emission center with different emissions due to the two main excited states of vacancies and the emissions due to excitation of $4f^1$ electron of Ce^{3+} .

In the case of UV excitation, PL emissions are mainly generated by grain surfaces and given the low energy of irradiation (3.6 eV), excited electrons can jump up to the bottom of CB. In the

case of X-ray excitation, the penetration of X-rays concerns the bulk of grains and due to the high energy polychromatic beam, excited electrons jump in higher energy levels, which modifies the transition conditions and can be at the origin of a complex quenching process.

The CT transitions $W5d \rightarrow O2p$ are fully quenched in our experiments because of the existence of different levels acting as trapping centers (Ce^{3+} , Sm^{3+} , defects).

The increasing red and IR emission intensities as Sm^{3+} content increases (and Ce^{3+} content decreases) are directly correlated to the presence of increasing number of defects (vacancies and distortions of oxygen polyhedra).

In addition to the thin emission peaks, a very large continuous emission covering the range 500 to 1000 nm is observed: this means that a large distribution of defects and local structural distortions exist in relation with the distortions observed in the above X-ray diffraction analysis due substitution and presence of vacancies and clusters Ce-V_O-W.

Table 3 reports the different PL emissions observed under UV and X-ray excitations and their attributions. **Fig. 11** is a schematic representation of the various transitions.

Table 3: Photoluminescence emissions under UV and X-ray excitations observed in the series $Ce_{2-x}Sm_x(WO_4)_3$. Transitions generated by presence of oxygen vacancies V^x , V° and $V^{\circ\circ}$.

UV excitation		X-ray excitation		Attributions
Wavelength (nm) Color	Energy (eV)	Wavelength (nm) Color	Energy (eV)	Transitions Ce^{3+} and Sm^{3+}
497.75 Green	2.49	No emission		Ce: ${}^2T_{2g} / {}^2D_{3/2} \rightarrow {}^2F_{7/2}$
524.61 Green	2.36	No emission		Ce: ${}^2T_{2g} / {}^2D_{5/2} \rightarrow {}^2F_{5/2}$
564.33 Yellow	2.19	563 Yellow	2.20	Sm: ${}^4G_{5/2} \rightarrow {}^6H_{5/2}$
600.50 Orange	2.064	599 Orange	2.070	Sm: ${}^4G_{5/2} \rightarrow {}^6H_{7/2}$
616.56 Orange	2.013	617 Orange	2.009	$V^{\circ\circ} \rightarrow V^\circ$
643.2 Red	1.927	644 Red	1.925	Sm: ${}^4G_{5/2} \rightarrow {}^6H_{9/2}$
667 Red	1.859	669 Red	1.853	$V^x \rightarrow V^x$
705 Dark red	1.759	705 Dark red	1.759	Sm: ${}^4G_{5/2} \rightarrow {}^6H_{11/2}$
900 Large IR	1.38	880 Large IR	1.41	$V^\circ \rightarrow$ holes O2p
950 Large IR	1.30	930 Large IR	1.33	$V^x \rightarrow$ holes O2p

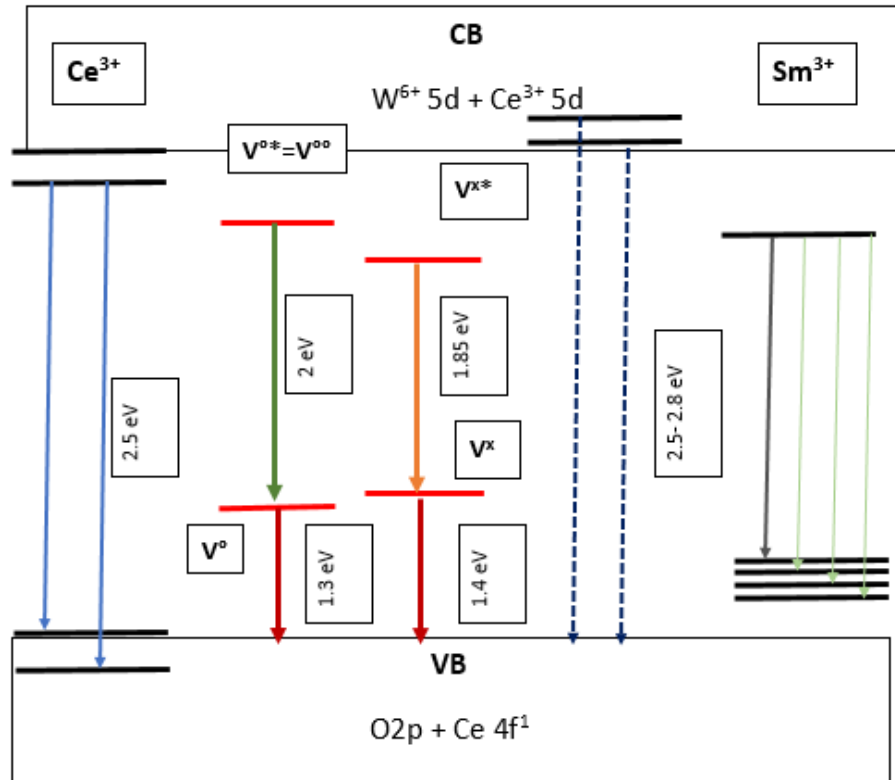


Fig. 11: Schematic representation of PL transitions in samarium cerium tungstates. Oxygen vacancies noted as V^x , V° , $V^{\circ\circ}$. VB = valence band with $O2p$ orbitals. CB = conduction band with $W5d$ and $Ce5d$ orbitals.

V- CONCLUSION.

The solid solution $Ce_{2-x}Sm_x(WO_4)_3$ with $x \leq 0.3$, obtained by coprecipitation followed by thermal treatment at $1000^\circ C$, is characterized by two PL emissions of Ce^{3+} observed only under UV excitation, and 4 PL emissions of Sm^{3+} observed under UV and X-ray excitations. Two additional transitions are observed under UV and X-ray excitations and are attributed to oxygen vacancy defects. As composition x increases, we observe a weak increase then a decrease of transition intensities linked to cerium, and a decrease of samarium PL emissions. The quenching effects observed in this material are probably amplified by the presence of defects increasing in number with x . The IR emission observed in the case of UV and X-ray excitations increases with x : this argues in favor of increase of the number of structural defects as x increases. To conclude, this series of solid solution presents different PL spectra under UV and X-ray excitations. The chromaticity diagram shows that the colors associated with PL responses vary with Sm composition and excitation energies (X-ray or UV). This offers the opportunity to develop materials with tunable PL. To better understand this complex behavior, now, we plan to study the solid solution in the composition range $0.3 < x \leq 2$.

ANNEX: atom coordinates for Rietveld calculations

The refined atom coordinates of T. Gressling et al. [13] were used to apply the Rietveld method to each X-ray diffraction pattern.

Ce	8f	0,3212	0,3734	0,4048
W(1)	4e	0.0	0,1184	0,25
W(2)	8f	0.1543	0,3547	0,0479
O(1)	8f	0.0129	0,0404	0,3874
O(2)	8f	0,1915	0,2056	0,2781
O(3)	8f	0,0717	0,3008	0,4620
O(4)	8f	0.2184	0,4265	0,1915
O(5)	8f	0.1383	0,5370	0,4402
O(6)	8f	0.3831	0.2858	0,0765

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